

**Thermo-Calc  
Software**

**Boosting the CALPHAD Modeling of Multi-  
component Systems by ab initio Calculations:  
Selected Case Studies**

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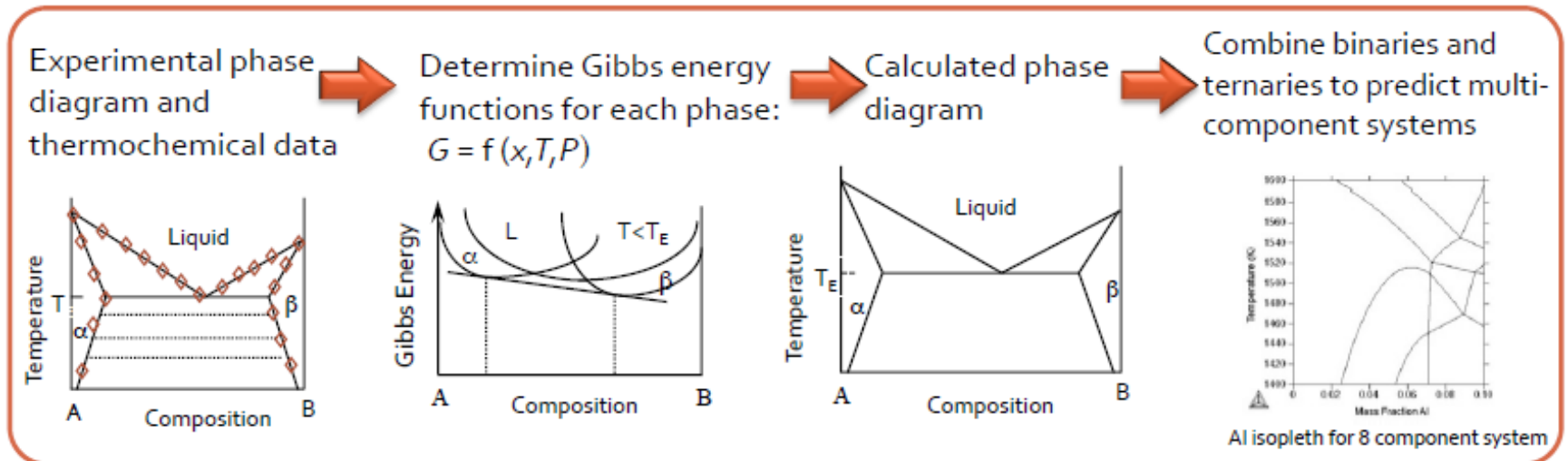
## Background and objectives

- Background: DFT emerged as a crucial source of input data to build **CALPHAD** thermodynamic “**databases**” for **multi-component compounds**.
- Challenge: The number of DFT calculations required to determine the parameters of models based on the standard **Compound Energy Formalism (CEF)** increases rapidly with the number of elements and sublattices.

# The CALPHAD thermodynamic modeling framework: the main idea

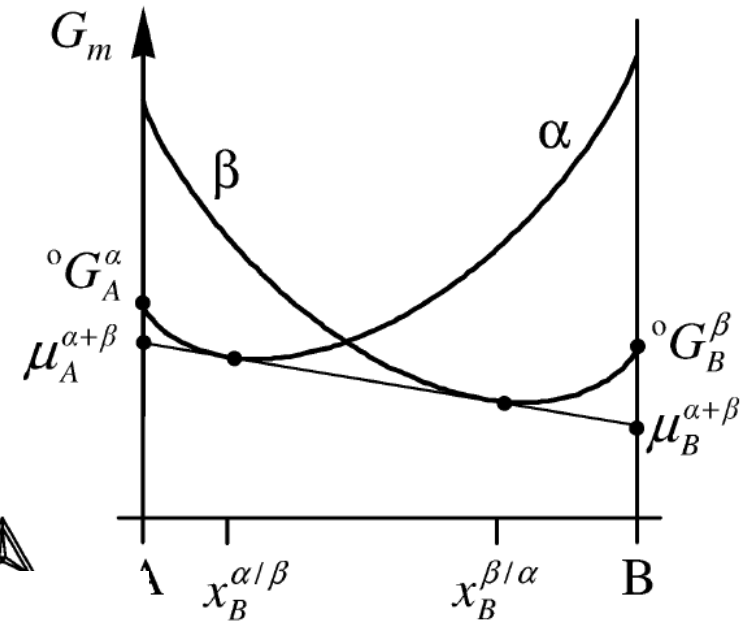
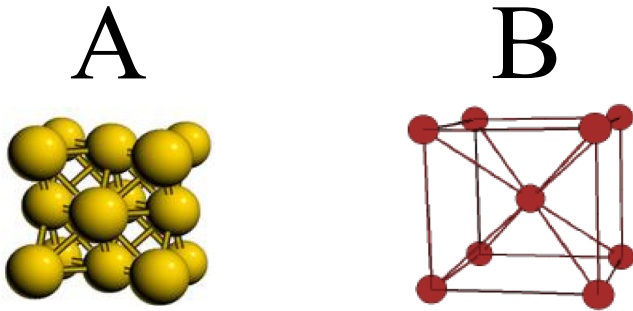
## CALPHAD (CALculation of PHase Diagrams)

- **Experimental** and **ab initio** data about the thermophysical data of an alloy are collected/generated.
- The thermophysical data are used to fit the **Gibbs energy functions** of the phases that the alloy exhibits.
- The Gibbs energy functions are applied to calculate thermodynamic equilibrium properties including phase diagrams



# A closer look at the CALPHAD method: phases with a single sublattice

The Gibbs free energy  $G_m$  of all the phases of an  $A_{1-x}B_x$  alloy is modeled across the full composition range



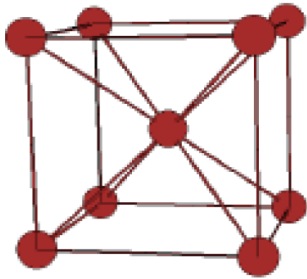
$$G_m(\{x_i\}, T) = {}^{\text{ref}}G + {}^{\text{id}}G + {}^{\text{E}}G$$

$$= \sum_i x_i {}^{\circ}G_i + RT \sum_i x_i \log(x_i) + {}^{\text{E}}G$$

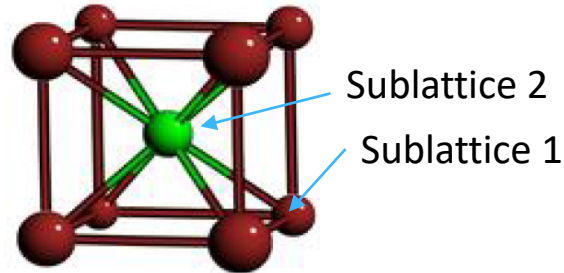
$${}^{\text{E}}G = \sum_i \sum_{j>i} x_i x_j \left[ \sum_{\nu} {}^{\nu}L_{ij} (x_i - x_j)^{\nu} \right]$$

# A closer look at the CALPHAD method: multi-sublattice phases

Single sublattice,  
e.g., BCC



Multiple sublattices,  
e.g., CsCl (B2) structure



Compound energy formalism to model the Gibbs energy

$$G_m(\{y_i^{(s)}\}, T) = {}^{\text{ref}}G + {}^{\text{id}}G + {}^{\text{E}}G$$

$${}^{\text{ref}}G = \sum_i \sum_j y_i^{(1)} y_j^{(2)} \circ G_{i:j}$$

$${}^{\text{id}}G = RT \sum_s a^{(s)} \sum_i y_i^{(s)} \log y_i^{(s)}$$

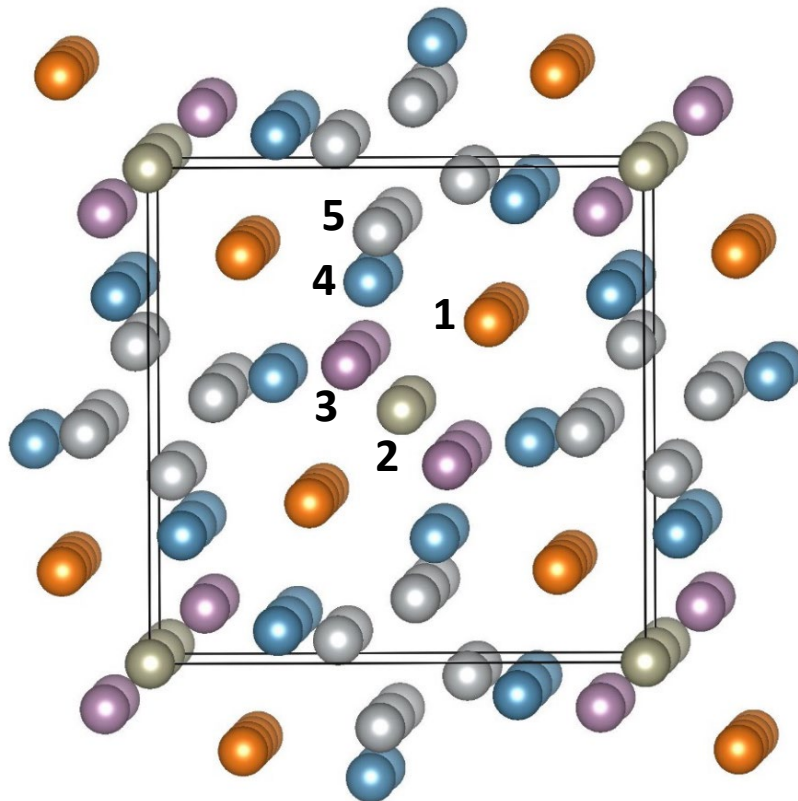
Example of binary  
end-members

$$\left[ \begin{array}{l} \circ G_{\text{Al:Al}} \\ \circ G_{\text{Ni:Ni}} \\ \circ G_{\text{Al:Ni}} \\ \circ G_{\text{Ni:Al}} \end{array} \right]$$

# Phases with numerous sublattices (e.g., five) are common, including in industrially relevant alloys!

Common multi-sublattice phases: topological closed-packed (TCP) phases, e.g.,  $\sigma$ -phase,  $\mu$ -phase,  $\chi$ -phase, C14, C15, etc.

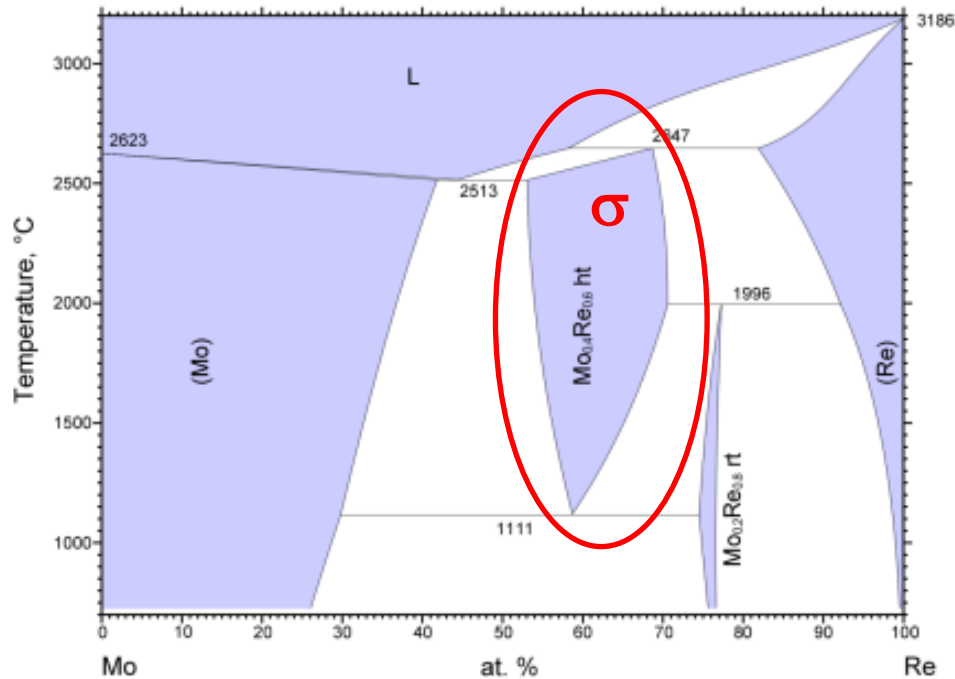
## $\sigma$ -phase structure



- 30 atoms in the unit cell
- 5 distinct lattice sites (Wyckoff positions)

# Example of binary alloy with a $\sigma$ -phase: Mo-Re

**Mo-Re**  
Experimental phase diagram



Number of binary sigma-phase end-members:  $2^5 = 32$

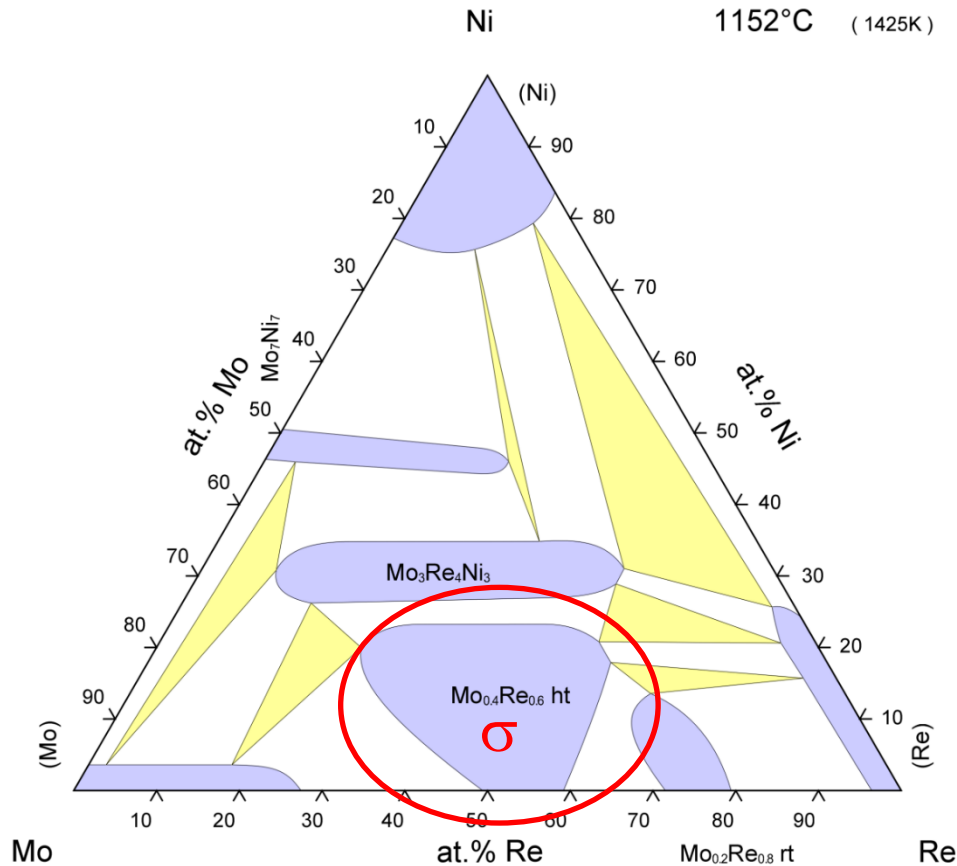
- $G_{\text{Re:Mo:Mo:Mo:Mo}}^{\sigma}$
- $G_{\text{Mo:Re:Mo:Mo:Mo}}^{\sigma}$
- $G_{\text{Re:Re:Mo:Mo:Mo}}^{\sigma}$
- $G_{\text{Mo:Mo:Mo:Re:Mo}}^{\sigma}$
- $G_{\text{Mo:Mo:Re:Mo:Mo}}^{\sigma}$
- .....

Approach to calculating the formation energies of the end-members:

**Density Functional Theory!**

# Example of ternary alloy with a $\sigma$ -phase: Mo-Ni-Re

Mo-Ni-Re phase diagram  
1425 K isothermal section



Number of binary  $\sigma$ -phase  
end-members:  $3^5 = 243$

- $G_{\text{Re:Mo:Ni:Mo:Mo}}^{\sigma}$
- $G_{\text{Mo:Re:Mo:Mo:Mo}}^{\sigma}$
- $G_{\text{Re:Re:Ni:Ni:Ni}}^{\sigma}$
- $G_{\text{Ni:Mo:Mo:Re:Ni}}^{\sigma}$
- $G_{\text{Mo:Mo:Re:Mo:Mo}}^{\sigma}$

.....



# Challenge for systematically describing the sigma phase in a real-world thermodynamic "database"

Elements in Thermo-Calc's TCNI9 thermodynamic database:



Al	Ar	B	C	Ca	Co	Cr	Cu	Fe	H
Hf	Mg	Mn	Mo	N	Nb	Ni	O	Pd	Pt
Re	Ru	S	Si	Ta	Ti	V	W	Y	Zr

## Some statistics about a typical database:

- Number of elements: **30**
- Number of binary (A-B) combinations:  $\sim 10^2$
- Number of ternary (A-B-C) combinations:  $\sim 10^3$

## Number of sigma end-members per element combination:

- Number of sigma **binary** end-members: **30**
- Number of sigma **ternary** end-members: **150**

## Number of DFT calculations needed to model the sigma phase in a typical database:

- Binary sigma phase:  $\sim 10^3$
- Number of ternary A-B-C combinations:  $\sim 10^5$

# The effective bond energy formalism (EBEF): the main idea

Dupin, Kattner, Sundman, Palumbo, and Fries  
J. Res. Natl. Inst. Stan. **123**, 123020 (2018)

- Expand the end-member energies into binary terms featuring only element pairs

→ Example of expansion of the energy for a **binary** end-member

$$\Delta G_{\text{Mo:Re:Mo:Mo:Mo}}^{\phi} = \Gamma_{\text{Mo:Re:***}} + \Gamma_{\text{*:Re:Mo:**}} + \Gamma_{\text{*:Re:**:Mo:*}} + \Gamma_{\text{*:Re:***:Mo}}$$

→ Example of expansion of the energy for a **ternary** end-member

$$\Delta G_{\text{Mo:Mo:Mo:Re:Ni}}^{\phi} = \Gamma_{\text{Mo:**:Re:*}} + \Gamma_{\text{Mo:***:Ni}} + \Gamma_{\text{*:Mo:**:Re:*}} + \Gamma_{\text{*:Mo:**:Ni}} + \Gamma_{\text{**:*:Mo:Re:*}} + \Gamma_{\text{**:*:Mo:**:Ni}} + \Gamma_{\text{***:Re:Ni}}$$

- **Interesting observation:** these expansions involve only binary terms.

# The effective bond energy formalism (EBEF): non-uniqueness of the solution and a remedy to this problem

Dupin, Kattner, Sundman, Palumbo, and Fries  
J. Res. Natl. Inst. Stan. **123** 123020 (2018)

→ Expand the end-member energies into binary terms featuring only element pairs

$$\Delta G_{i:j:k:l:m}^{\phi} = \sum_i \sum_{j \neq i} \left( y_i^{(1)} y_j^{(2)} \Gamma_{i:j:***} + y_i^{(1)} y_j^{(3)} \Gamma_{i:**:***} + y_i^{(1)} y_j^{(4)} \Gamma_{i:***:***} + y_i^{(1)} y_j^{(5)} \Gamma_{i:***:**} + y_i^{(2)} y_j^{(3)} \Gamma_{*:i:j:***} + y_i^{(2)} y_j^{(4)} \Gamma_{*:i:**:***} + y_i^{(2)} y_j^{(5)} \Gamma_{*:i:***:**} + y_i^{(3)} y_j^{(4)} \Gamma_{**:*:i:j:***} + y_i^{(3)} y_j^{(5)} \Gamma_{**:*:i:**:***} + y_i^{(4)} y_j^{(5)} \Gamma_{***:**:i:j} \right)$$

The determination of the EBEF parameters is a linear regression problem:

$$y_i^{(s_1)} y_j^{(s_2)} \longrightarrow x_p$$

$$\Gamma_{i:j:***} \longrightarrow \beta_p$$

$$\Delta G_{i:j:k:l:m}^{\phi} \longrightarrow y_e$$

$$\mathbf{y} = \mathbf{X}\beta$$

$$\hat{\beta} = (\mathbf{X}^T \mathbf{X})^{-1} \mathbf{X}^T \mathbf{y}$$

**Warning:**  $\det(\mathbf{X}^T \mathbf{X}) = 0$

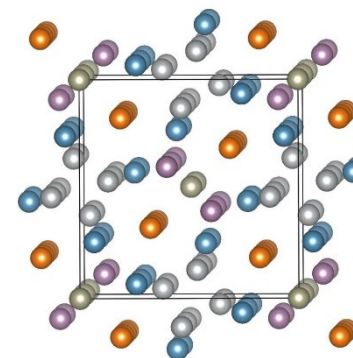
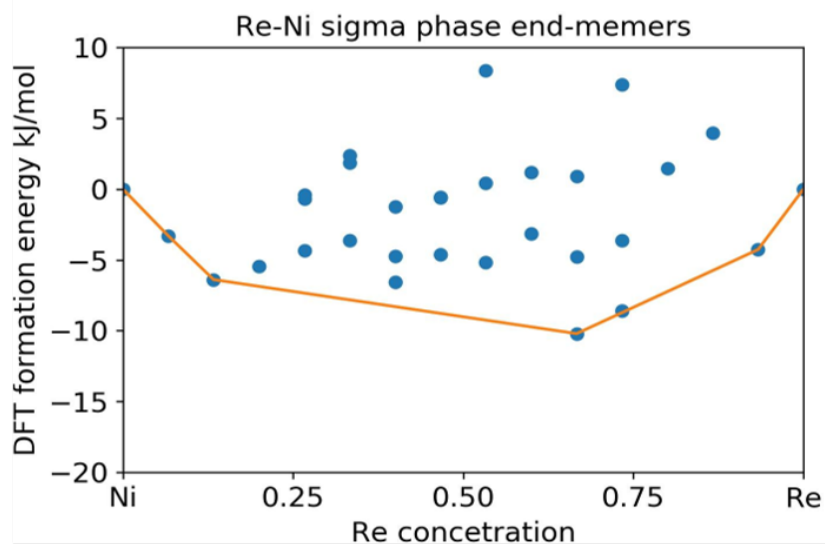
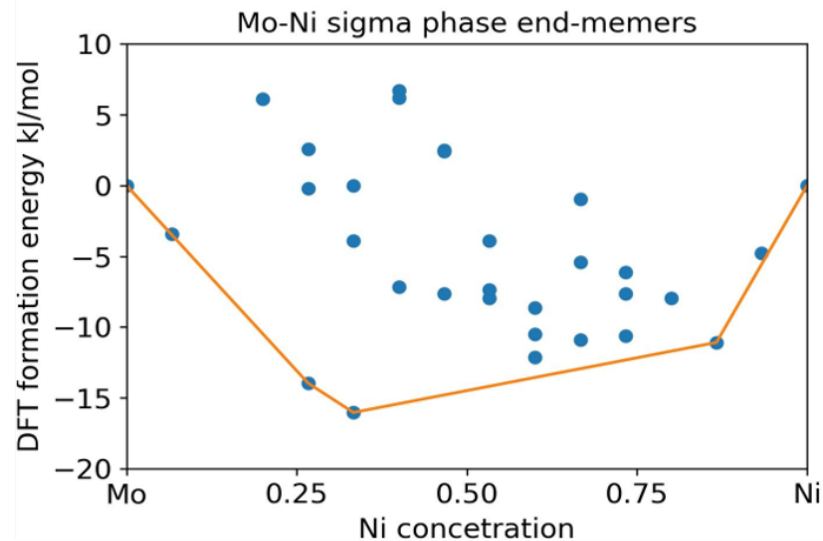
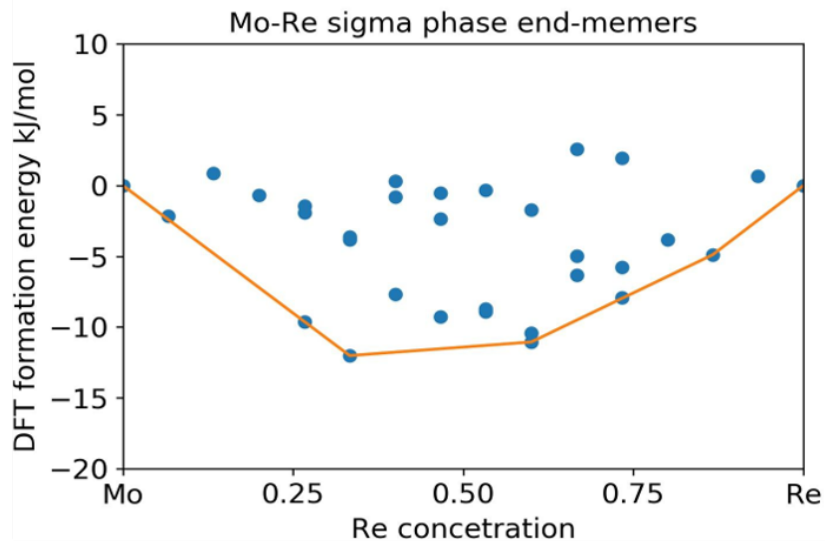
hence, there are infinite solutions

$$\hat{\beta}_{\text{SVD}} = \mathbf{V}\Sigma^{-1}\mathbf{U}^T \mathbf{y}$$

The **singular value decomposition (SVD)** gives the **minimum norm solution** which is **unique**

# Mo-Ni-Re: DFT convex hulls for the binaries

- DFT formation energies of the binary sigma end-members.
- The references are the total energies of the elements in the sigma phase.

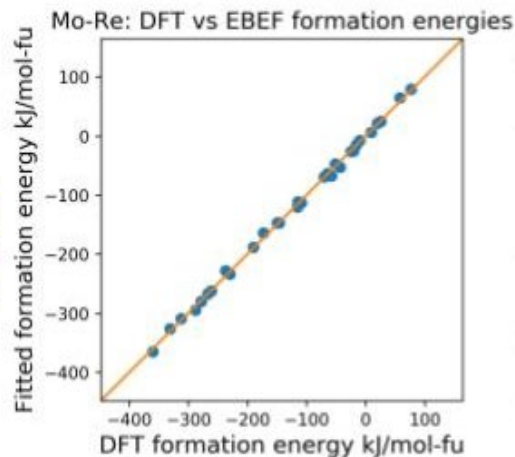


DFT total energies for the sigma endmembers from Yaqoob, Inorganic Chemistry **51**, 3071 (2012)

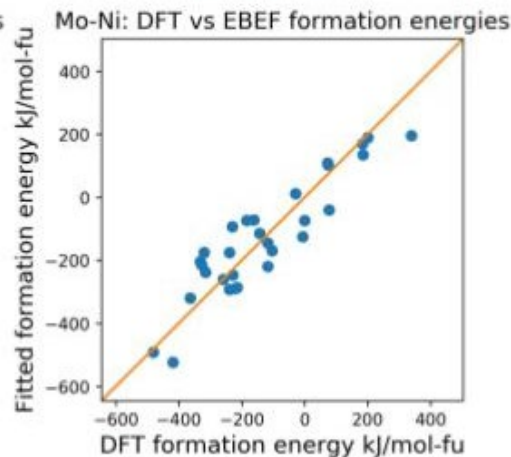
# The EBEF applied to Mo-Ni-Re: performance of the proposed implementation

Present  
work

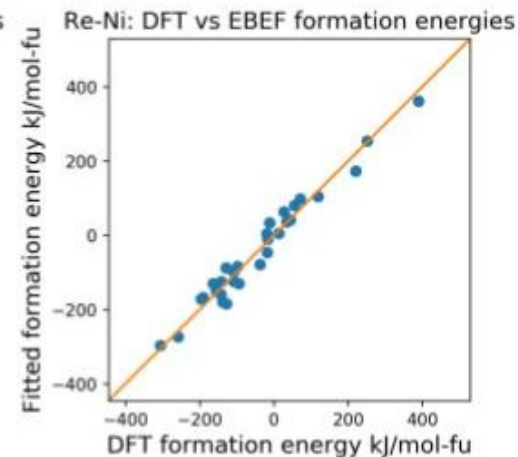
## Mo-Re



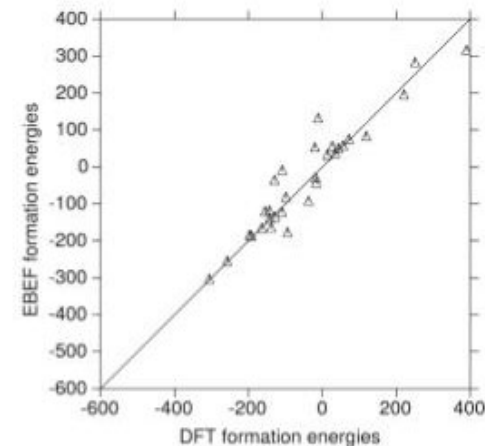
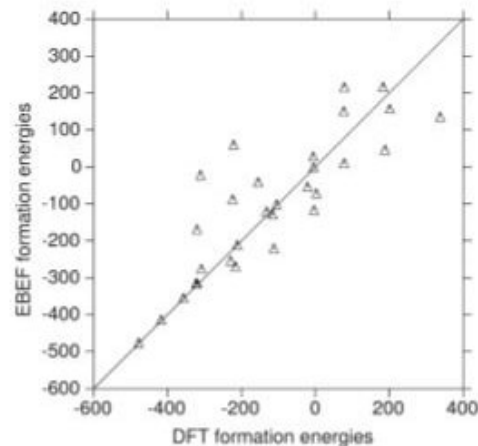
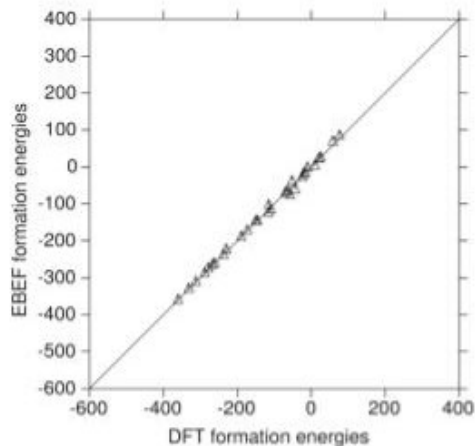
## Mo-Ni



## Re-Ni

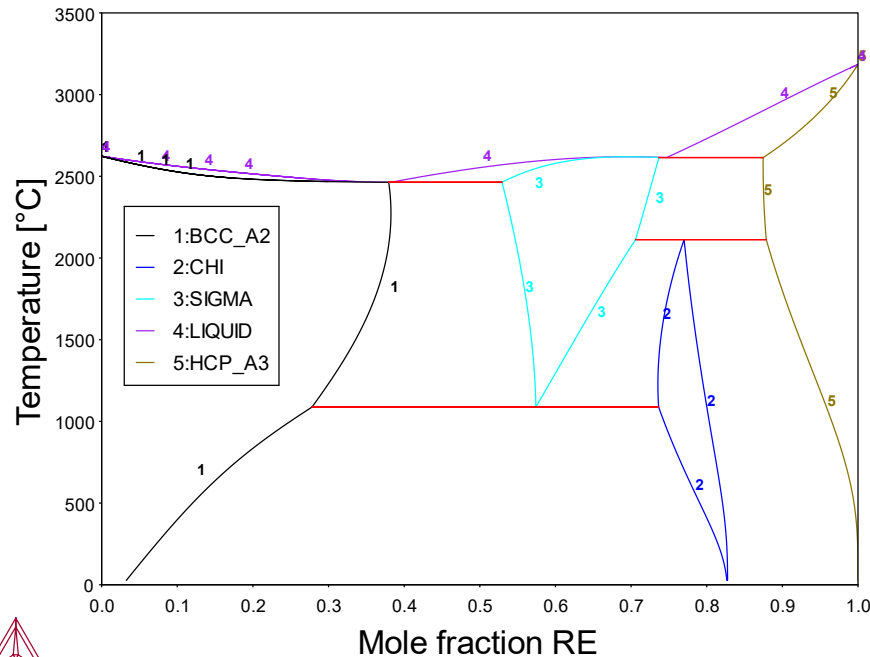


Dupin et al.



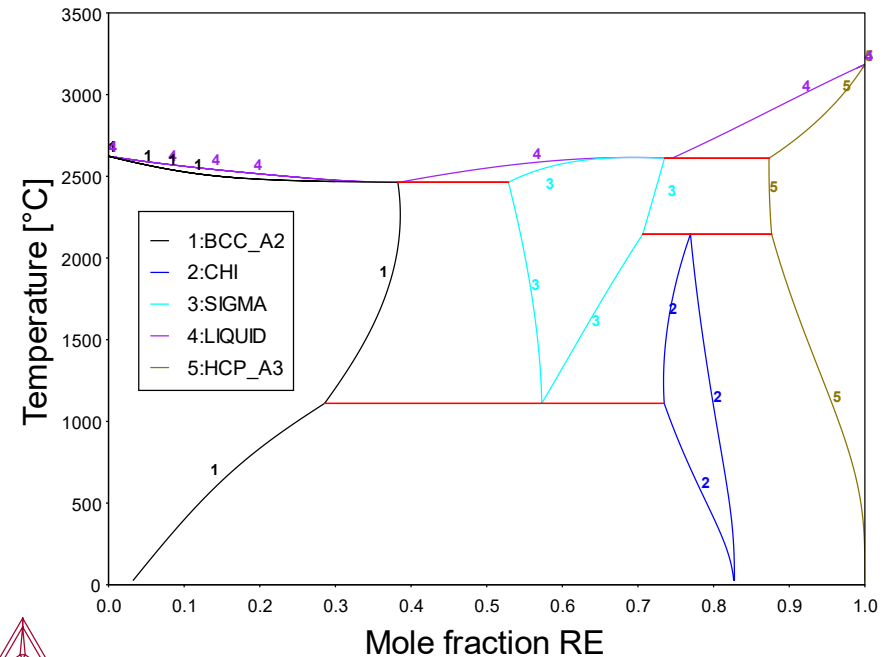
## Published EBEF parameters

2019.09.04.22.42.26  
USER: MO, RE  
P=1E5, N=1



## EBEF parameters obtained here

2019.09.04.22.42.39  
USER: MO, RE  
P=1E5, N=1

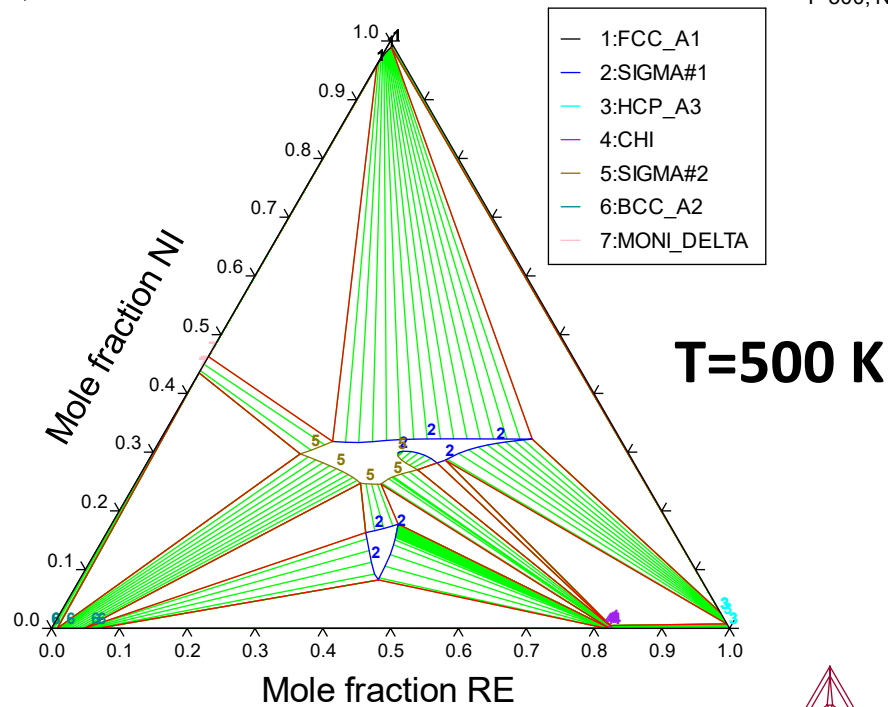


DFT total energies for the sigma endmembers from  
Yaqoob, Inorganic Chemistry 51, 3071 (2012)

# EBEF description of Mo-Ni-Re: performance of the proposed implementation

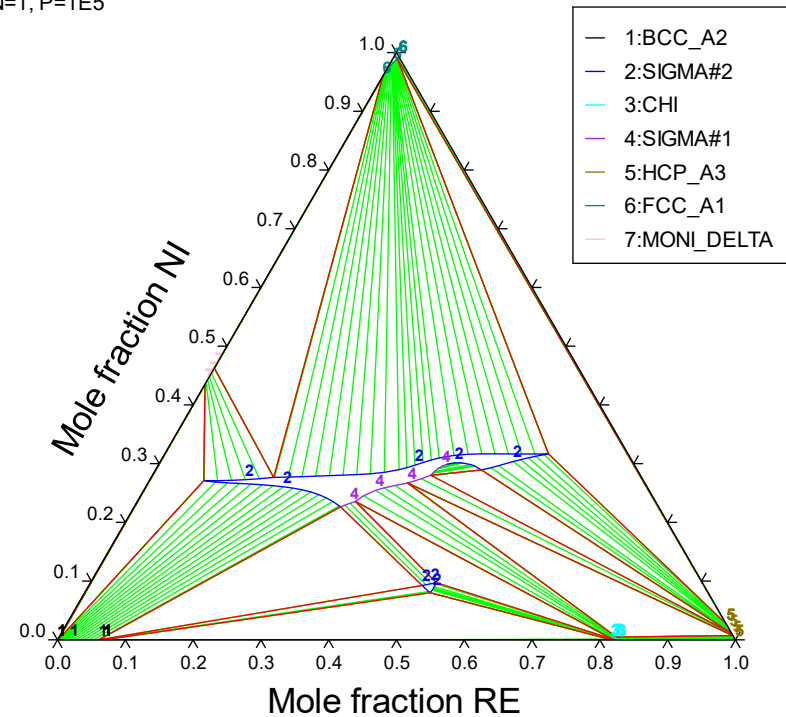
## Published EBEF parameters

2019.09.03.22.00.37  
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T=500, N=1, P=1E5



## EBEF parameters obtained here

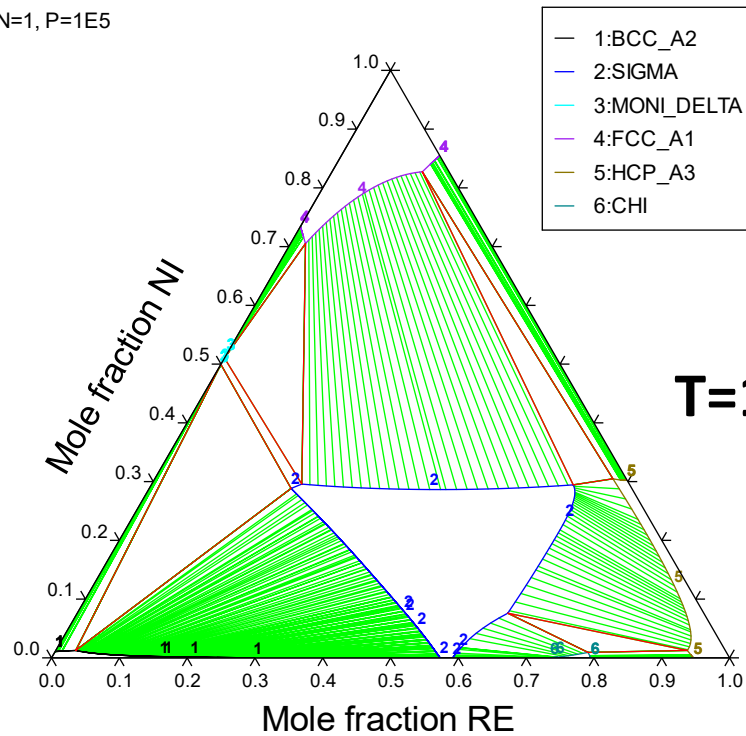
2019.09.03.21.57.04  
USER: MO, NI, RE  
T=500, N=1, P=1E5



# EBEF description of Mo-Ni-Re: performance of the proposed implementation

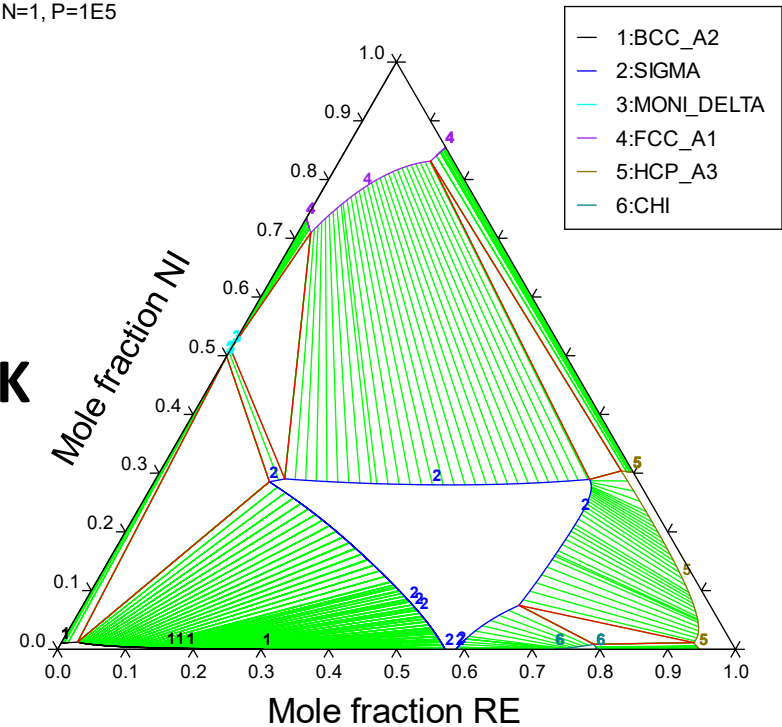
## Published EBEF parameters

2019.09.03.22.01.48  
USER: MO, NI, RE  
T=1500, N=1, P=1E5



## EBEF parameters obtained here

2019.09.03.21.57.58  
USER: MO, NI, RE  
T=1500, N=1, P=1E5

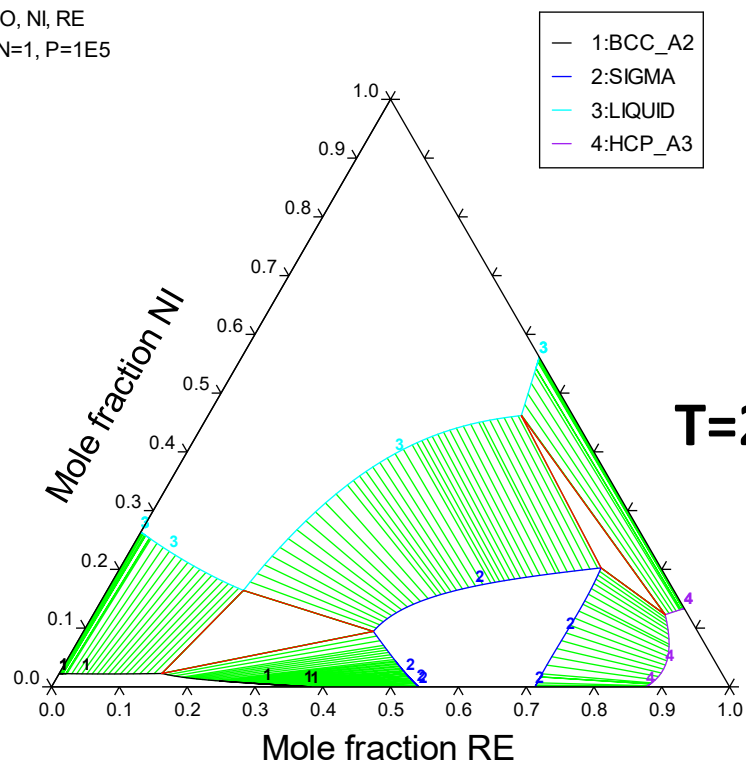




# EBEF description of Mo-Ni-Re: performance of the proposed implementation

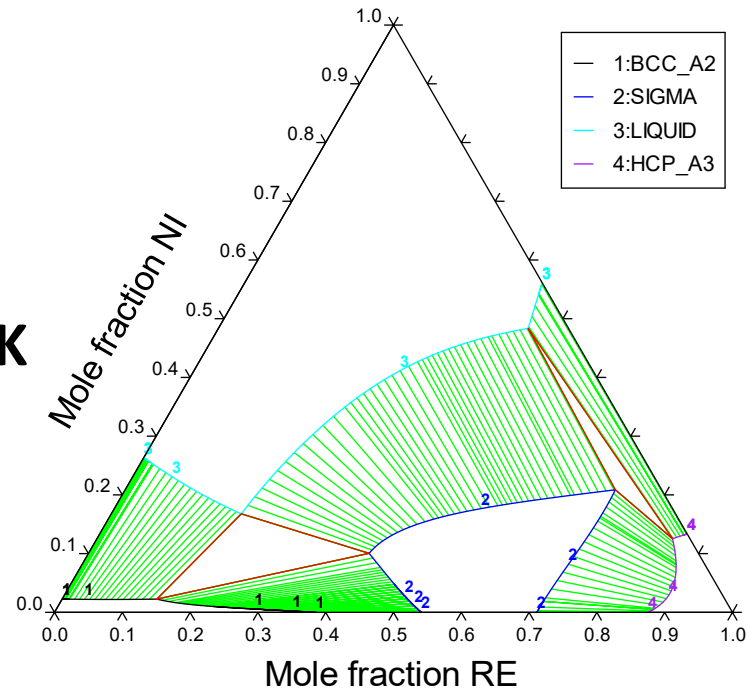
## Published EBEF parameters

2019.09.03.22.02.16  
USER: MO, NI, RE  
T=2500, N=1, P=1E5



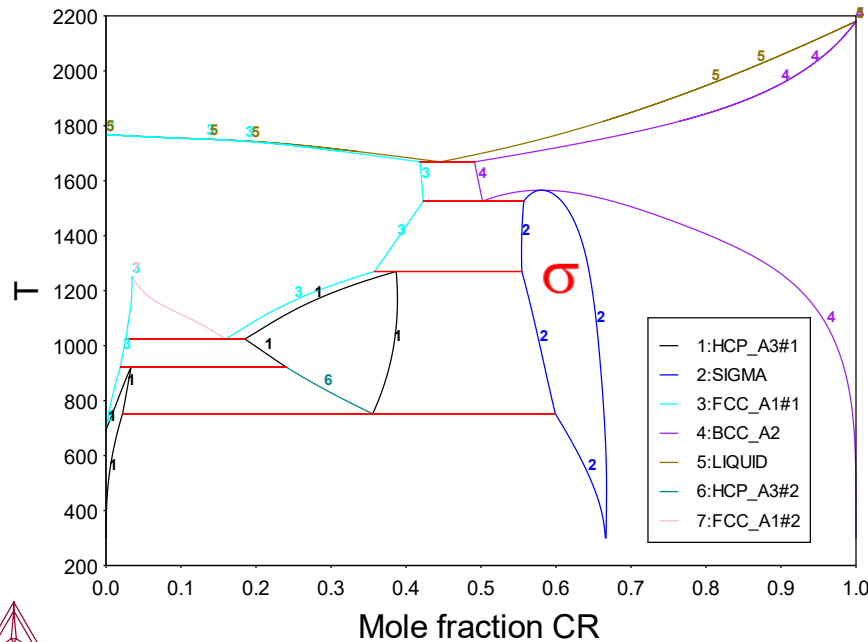
## EBEF parameters obtained here

2019.09.03.21.58.11  
USER: MO, NI, RE  
T=2500, N=1, P=1E5



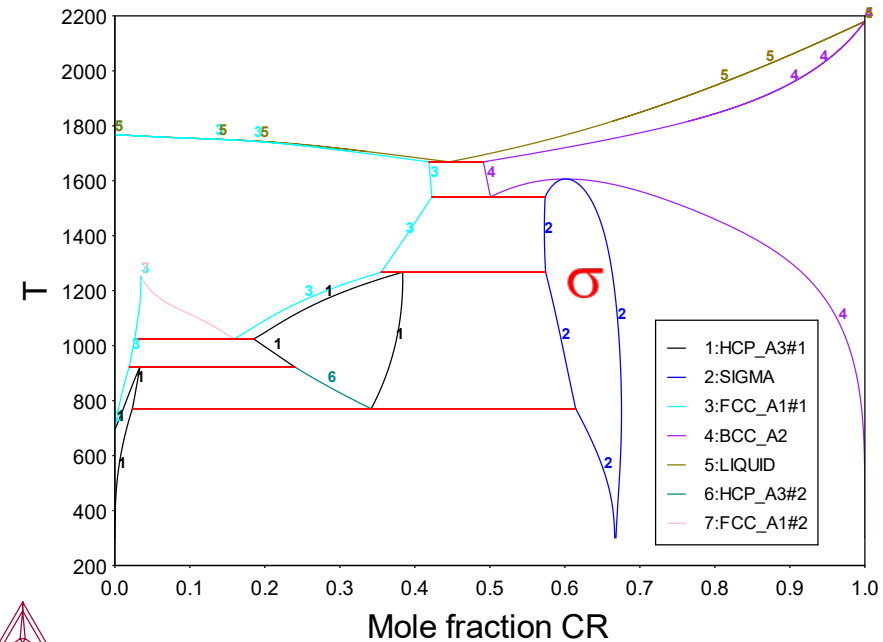
## DFT-based CEF

2019.09.04.07.41.13  
USER: CO, CR  
P=1E5, N=1



## DFT-based EBEF

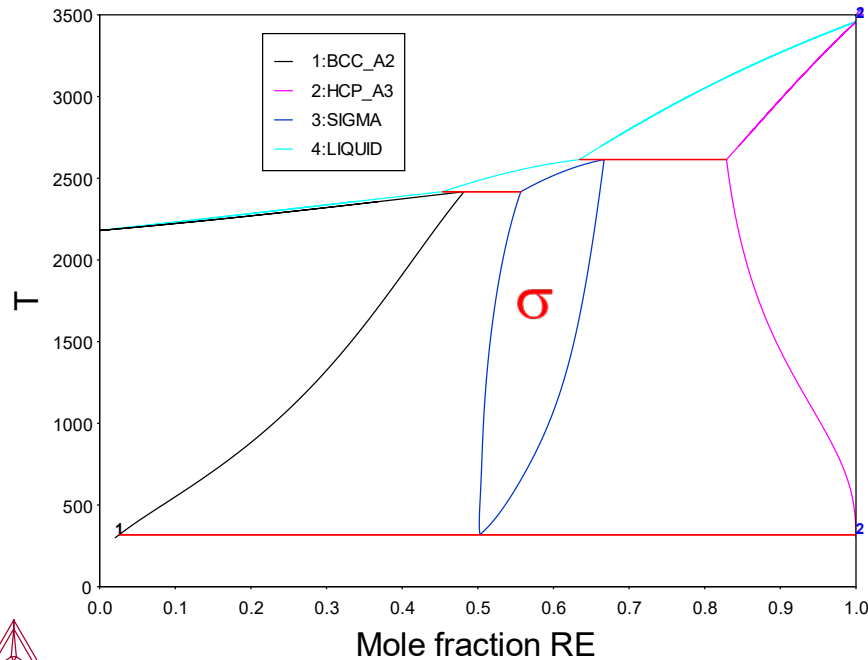
2019.09.04.07.41.49  
USER: CO, CR  
P=1E5, N=1



Original thermodynamic database and DFT data for the sigma endmembers from Li, Mao, Korzhavyi, and Selleby, CALPHAD **34**, 495 (2010)

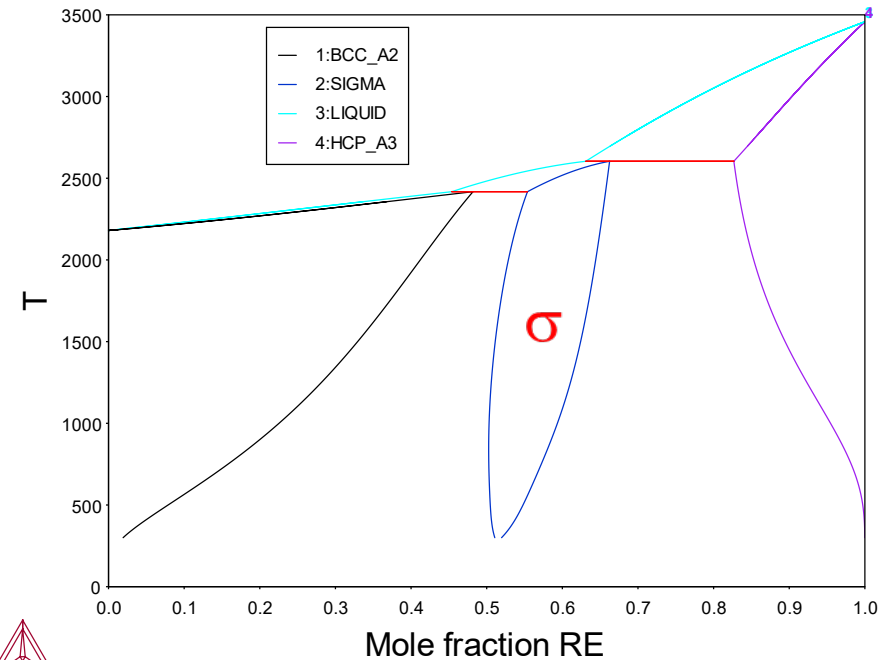
## DFT-based CEF

2019.09.04.08.40.53  
USER: CR, RE  
P=1E5, N=1



## DFT-based EBEF

2019.09.04.08.33.31  
USER: CR, RE  
P=1E5, N=1



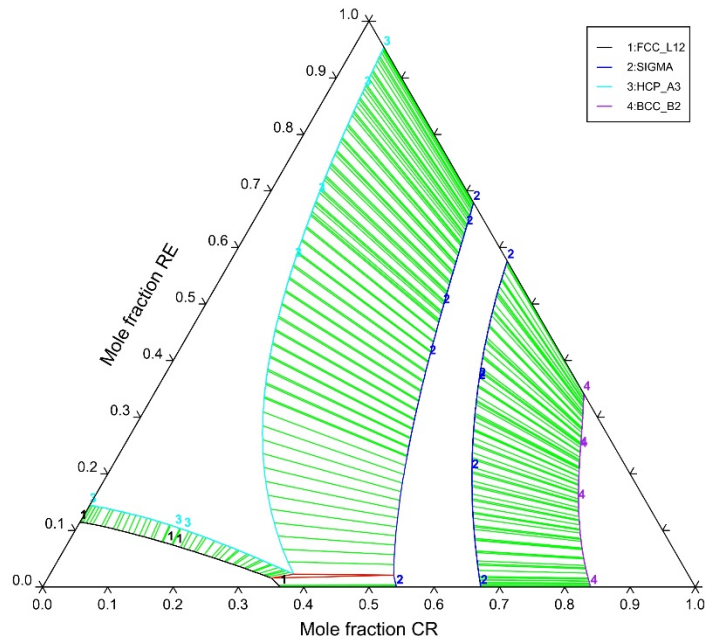
DFT data for the sigma endmembers from Palumbo et al., CALPHAD **34**, 495 (2010)

# Co-Cr-Re: Compound energy formalism vs Effective Bond Energy Formalism

## TCNI8

CO-CR-RE at T=1273.15 K

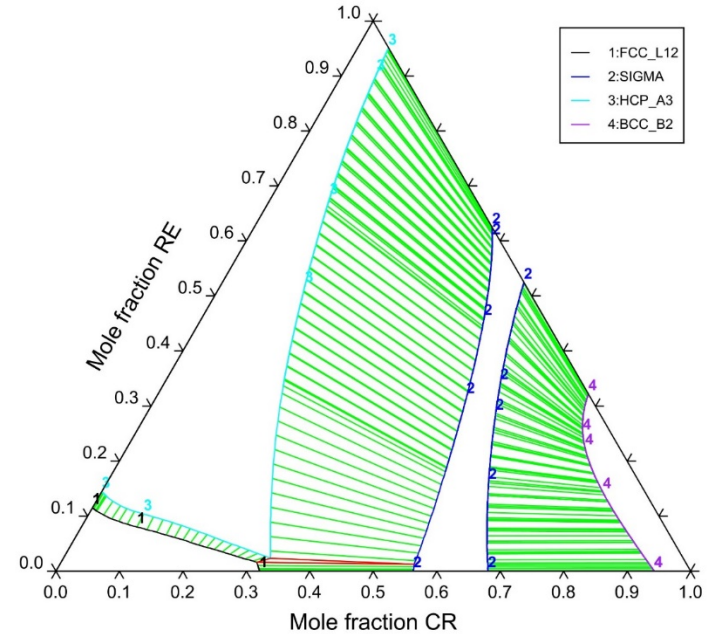
2018.10.17.17.00.53  
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T=1273.15, P=1E5, N=1



## DFT-based EBEF

CO-CR-RE at T=1273.15 K

2018.10.20.12.04.35  
USER: CO, CR, RE  
T=1273.15, P=1E5, N=1



## Where we are and outlook....

- Presented an assessment of the EBEF approach.
- Performed further tests of the performance of the EBEF on selected topologically close-packed phases.
- EBEF does emerge as a way to obtain a reasonable description of a ternary multi-sublattice phase at the cost of DFT calculations for the binary end-members.
- The fitting algorithm that we propose for the EBEs allows to deal with the non-uniqueness of the solution according